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## **Organic Compound/Metal Oxide Hybrid Thin Film Prepared by the Liquid Phase Deposition Method**

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Organic compound/metal oxide composite thin films have been prepared by the liquid-phase deposition method in one-step deposition technique. Organic dyes were added to the metal fluoro-complexes aqueous solution in order to entrap these dyes within the growing thin films of metal oxides, yielding organic dye/metal oxide hybrid materials. According to Raman spectroscopy and X-ray photoelectron spectroscopy, the formation of the materials in the case of cationic dyes can be explained by electrostatic interaction between negative charge density at the fluorinated surface of metal oxides and the cationic dyes. UV-Vis results indicate that stilbazo and pyrocatechol violet (catecholate dye molecules) form a charge transfer complex with metal oxides through the catechol moiety with a bidentate linkage.

### **Introduction**

Hybrid organic-inorganic materials have received growing attention over the past 10 years, in particular because they are attractive candidates for optical devices, catalysts, sensor coatings etc. (1). The most common method of preparation of these materials is the insertion of organic molecules into the galleries or inorganic lattices, including 3-D frameworks, 2-D layers host, 1-D tunnel hosts, etc. (2,3). In general, the synthesis of organic/inorganic hybrid materials is carried out in two or more successive steps including the preparation of metal oxide, surface modification due to the need of high temperature treatment, etc. (4,5).

We have recently developed and promoted a novel aqueous solution-based process to prepare metal oxide thin films, the so-called liquid phase deposition (LPD) method (6-9). The advantage of the LPD method is the ability to synthesize hybrid materials under an ambient condition without any post-annealing process. Based on this technique, organic/inorganic hybrid thin films such as alkyl sulfate and alkylbenzene sulfonate surfactants/TiO<sub>2</sub> hybrid films and protein-templated organic/inorganic hybrid materials have been developed (10,11). However, to-date, there are only a few investigation on the molecular structures of organic compound which can be incorporated into metal oxide films. The effect of molecular structures on the deposition rate of metal oxide has not yet been reported.

In this study, we have prepared organic dye/metal oxide hybrid thin films by LPD method, and investigated the structural properties of organic compound incorporated into metal oxide films. The synthesized materials have been characterized by Raman

spectroscopy, XPS, UV–vis spectroscopy, SEM and ICP-AES.

## Experimental

### Materials

Soda-lime glass was used as the substrate.  $\text{H}_3\text{BO}_3$  and Al metal were employed as the  $\text{F}^-$  scavenger. The following complex material were used as parent solutions: titanium oxide  $\{(\text{NH}_4)_2\text{TiF}_6\}$ , silicon oxide  $\{(\text{NH}_4)_2\text{SiF}_6\}$ , zirconium oxide  $\{\text{H}_2\text{ZrF}_6\}$ . The organic dyes consisted of: methylene blue (MB), neutral red (NR), rhodamine B (RB), Malachite green (MG), eosin Y (EY), stilbazo, pyrocatechol violet (PCV) and aluminon (Figure 1). Stilbazo and PCV are the catecholate dye molecules.

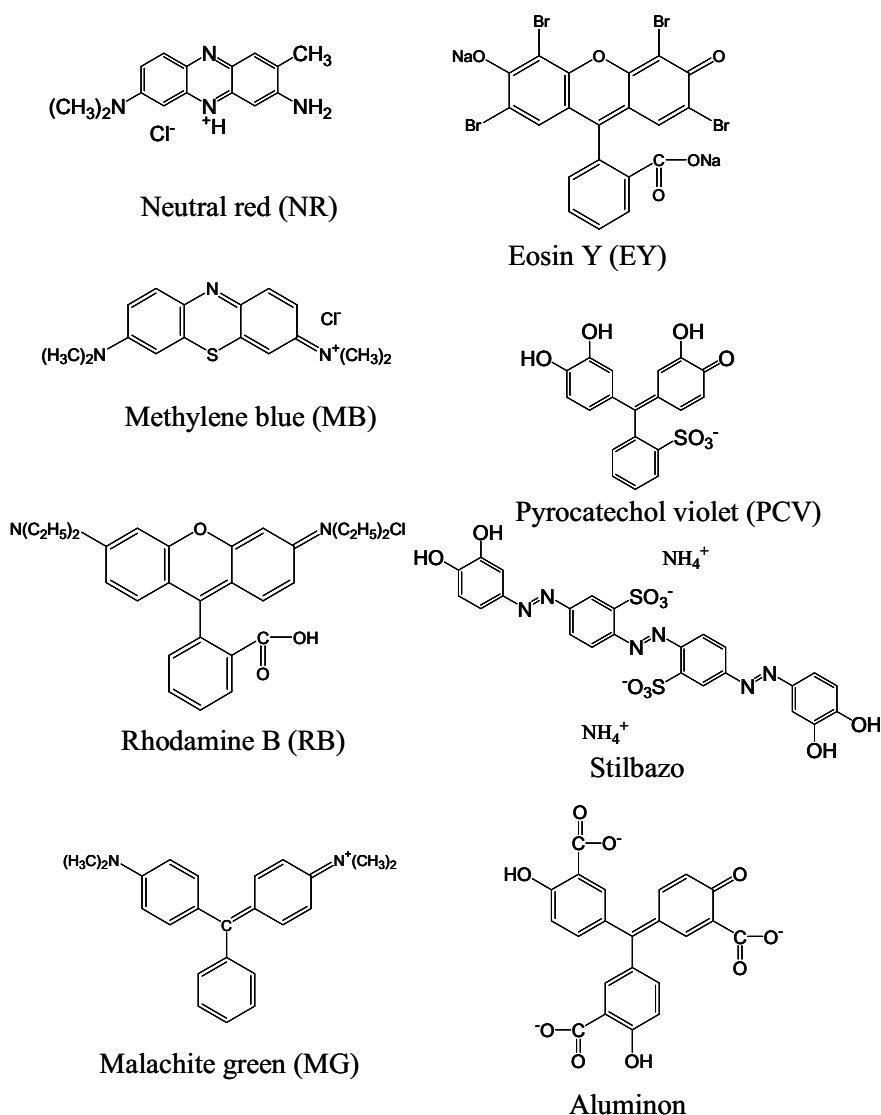


Figure 1. The structures of organic dyes.

### Sample preparation

The deposition of organic dye/metal oxide composite thin film was carried out in one step as follows. The substrates were immersed into the treatment solution containing metal-fluoro complex, the  $F^-$  scavenger and organic dye aqueous solution. These solutions were mixed at various compositions and used as the treatment solution. For each system, the following composition was used: titanium oxide system, 10 mol/dm<sup>3</sup> [(NH<sub>4</sub>)<sub>2</sub>TiF<sub>6</sub>] and 0.20 mol/dm<sup>3</sup> H<sub>3</sub>BO<sub>3</sub>; silicon oxide system, 0.10 mol/dm<sup>3</sup> [(NH<sub>4</sub>)<sub>2</sub>SiF<sub>6</sub>] and 0.20 mol/dm<sup>3</sup> H<sub>3</sub>BO<sub>3</sub>; zirconium oxide system, 50 mmol/dm<sup>3</sup> [H<sub>2</sub>ZrF<sub>6</sub>] and 2 x 12 cm<sup>-2</sup> Al metal. The organic dyes were dissolved in the treatment solution at a concentration of 0.0–1.0 mmol/dm<sup>3</sup>.

After degreasing and washing ultrasonically, the substrates were immersed into the treatment solution and suspended therein vertically. The reaction varied from: 2–52 h and the temperature from: 30–60°C. The substrates were then removed from the treatment solution, washed with distilled water and dried at ambient temperature.

### Sample characterization

The optical transmission and scattering measurements of the thin films were performed using UV-Vis Spectrophotometer (JASCO V 7200). The surface morphologies of the films were observed by field emission scanning electron microscope (FE-SEM; JEOL JEM-6335F). In order to prevent the surface of samples from electron charging up, the samples were coated using carbon coater (40FM; Meiwa Shoji Co. Ltd.). The amount of chemical elements contained in the deposited films was measured by inductively coupled plasma atomic emission spectrometry (ICP-AES; HORIBA Ltd., ULTIMA 2000). The deposited films on a substrate were dissolved into 20 ml of 3.78 mol dm<sup>-3</sup> HCl. The crystalline structures were investigated using X-ray diffractometer (Rigaku RINT-TTR/S2) equipped with a scintillation detector, and a rotating Cu anode operating radiation at 50 kV and 300 mA. Using parallel beam optics formed by a multilayered mirror (Rigaku, Cross Beam Optics attachment), asymmetric 2 $\theta$  scans with a fixed small incident angle were carried out. Raman spectra were measured with Horiba T-64000 excited by 532 nm SHG of Nd-YVO<sub>4</sub> laser. Infrared measurement were performed by means of FT-IR 615 type spectrophotometer (JASCO), coupled with a diffuse reflectance attachment DR-600B (JASCO). The resolution was 2 cm<sup>-1</sup>. Chemical states of F comprising into the films were analyzed with X-ray photoelectron spectroscopy (XPS; JEOL JPS-9010MC) with AlK $\alpha$  as the X-ray source (1486.6 eV) and charge neutralizer. Calibration for spectra was performed by taking the C 1s electron peak ( $E_b$  = 284.6 eV) as internal reference.

## **Results and discussion**

### Incorporation of organic compound into metal oxide thin film

Using the LPD process, from each treatment solution containing organic dye, metal oxide thin film was deposited on the substrate. The deposited films were observed with naked eyes from which, two categories of samples could be distinguished as shown in Table 1: (i) the circle mark which indicate colored thin film, (ii) samples marked with

**Table 1.** Organic dye – metal oxide composite thin film.

|                         | MB | NR | RB | MG | EY | Stilbazo | PCV | Aluminon |
|-------------------------|----|----|----|----|----|----------|-----|----------|
| TiO <sub>2</sub> system | ○  | ○  | ×  | ×  | ×  | ○        | ○   | ○        |
| SiO <sub>2</sub> system | ○  | ○  | ○  | ○  | ×  | ×        | ×   | ×        |
| ZrO <sub>2</sub> system | ○  | ○  | ×  | ×  | ×  | ○        | ○   |          |

○ ; Organic dye/metal oxide composite thin film  
 × ; non deposition or no colored

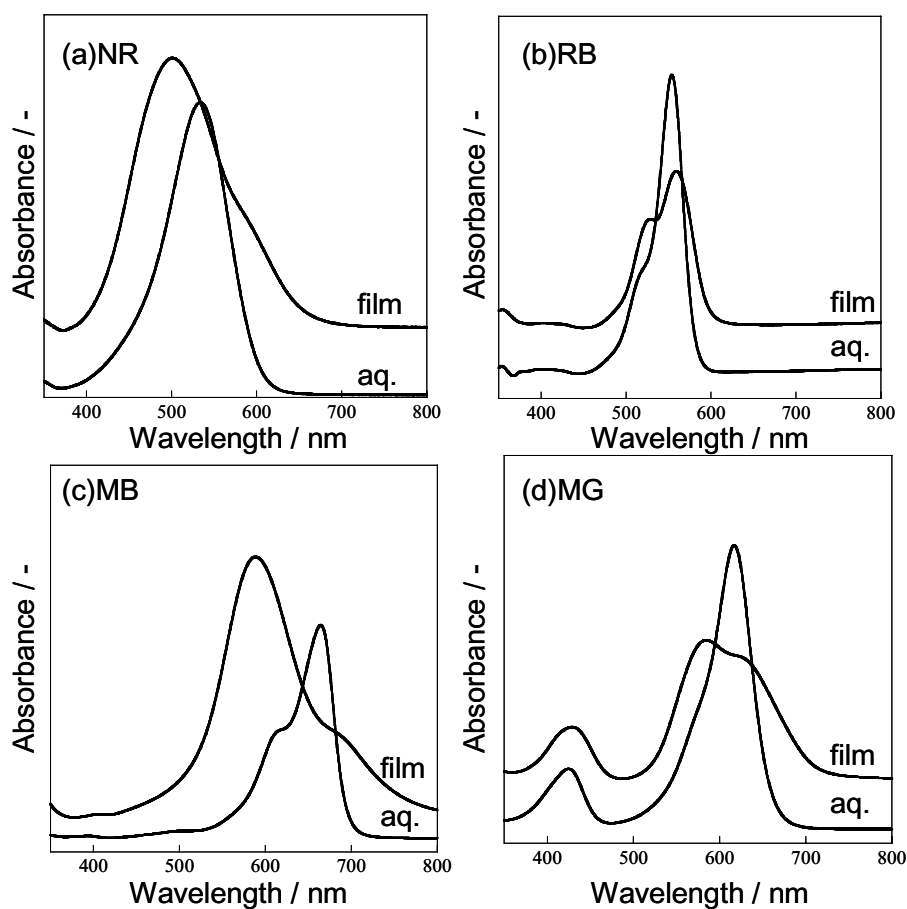
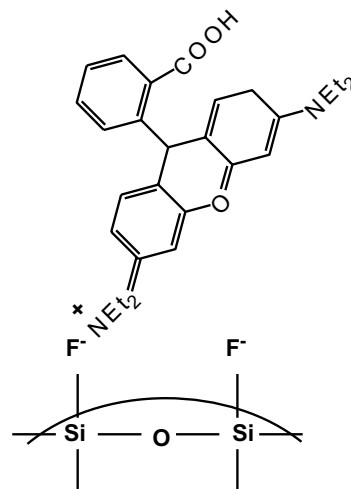


Figure 2. Absorption spectra of: (a) NR in aqueous solution and NR/SiO<sub>2</sub> film, (b) RB in aqueous solution and RB/SiO<sub>2</sub> film, (c) MB in aqueous solution and MB/SiO<sub>2</sub> film and (d) MG in aqueous solution and MG/SiO<sub>2</sub> film.

cross sign showing no colored or no deposition. In  $\text{SiO}_2$  system, cationic dyes (MB, NR, RB and MG)/ $\text{SiO}_2$  colored films were obtained whereas anionic dyes (EY, stilbazo, PCV and aluminon)/ $\text{SiO}_2$  films were colorless. On the other hand, in  $\text{TiO}_2$  and  $\text{ZrO}_2$  systems, hybrid films of cationic MB and NR were obtained, however hybrid films of cationic RB and MG did not show any color.  $\text{TiO}_2$  prepared by LPD process present a high degree of surface fluorination ( $\equiv\text{Ti-F}$ ) (12,13). The fluoride chemisorption greatly reduces the positive surface charge on  $\text{TiO}_2$  by replacing  $\equiv\text{Ti-OH}_2^+$  by  $\equiv\text{Ti-F}$  species (14). Due to the negative surface charge on  $\text{SiO}_2$ ,  $\text{TiO}_2$  and  $\text{ZrO}_2$ , hybrid films of cationic dyes were formed. In  $\text{SiO}_2$  system, colored hybrid films of cationic RB and MG could be fabricated because the electronegativity of Si is higher than that of Ti and Zr. As a result, fluorine adsorbs more easily onto Si than on Ti and Zr. In other words, the surface charge on  $\text{SiO}_2$  is more negative than on  $\text{TiO}_2$  and  $\text{ZrO}_2$ . Figure 2 shows the absorption spectra of free organic dyes in aqueous solution and organic dye/ $\text{SiO}_2$  hybrid films. The shift of the absorption maximum upon incorporation is stronger in MB and NR than in RB and MG. This indicates that the interaction to  $\text{SiO}_2$  is larger in MB and NR than in RB and MG. The cationic groups in RB and MG have weaker interaction due to steric hindrance of two ethyl groups (as illustrated in Scheme 1). Therefore, hybrid films of cationic RB and MG did not show any color in  $\text{TiO}_2$  and  $\text{ZrO}_2$  systems.



Scheme 1. Incorporation model of RB into  $\text{SiO}_2$ .

The formation of stilbazo/ $\text{TiO}_2$  hybrid film as a function of the reaction time can be monitored by following the evolution of the absorption spectra as shown in Figure 3. The intensity of absorption increased as the reaction time passed. This result indicates that stilbazo not only adsorbed on  $\text{TiO}_2$  surface, but also incorporated into the deposited film. Therefore, by using the LPD method, the organic compounds can be incorporated into various oxide films such as  $\text{TiO}_2$ ,  $\text{SiO}_2$  and  $\text{ZrO}_2$ .

Figure 4 shows Raman spectra of free MB in aqueous solution and MB/ $\text{TiO}_2$  hybrid film. Characteristic peaks of MB at  $\sim 1600\text{ cm}^{-1}$  (ring stretch) and

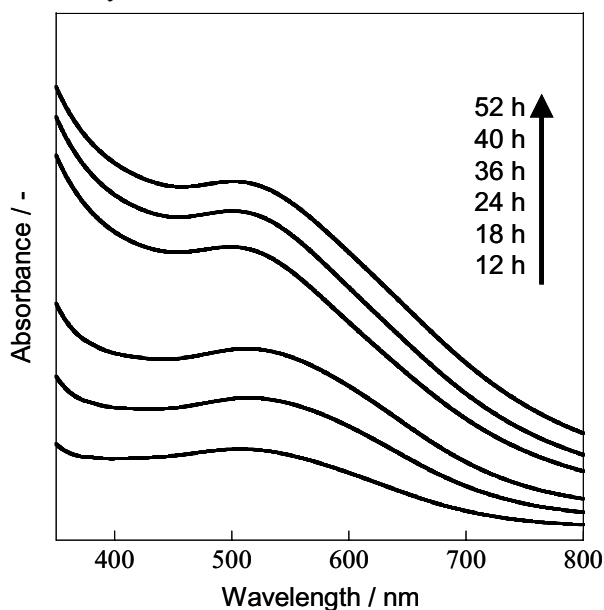


Figure 3. Absorption spectra of stilbazo/ $\text{TiO}_2$  film as a function of the reaction time.

$\sim 1390\text{ cm}^{-1}$  (C–N symmetric stretch) appear in the MB/TiO<sub>2</sub> hybrid film. CN ring vibration shifted toward lower wavenumber from  $1400\text{ cm}^{-1}$  to  $1390\text{ cm}^{-1}$  upon incorporation. This shift indicates an interaction of the MB with TiO<sub>2</sub>. It is thought therefore, that the electrostatic interaction between negative density charge at the TiO<sub>2</sub> fluorinated surface generated during the LPD process and the cationic MB ( $\text{Ti-F}^{\delta-}::\text{MB}^+$ ) is responsible of MB incorporation (15).

The XPS measurement was further employed to analyze fluorine of the surface of MB/TiO<sub>2</sub> hybrid film. The XPS technique provides a good tool for the study of interaction modes of MB with the TiO<sub>2</sub>. Figure 5 shows the F 1s XPS

spectra of MB/TiO<sub>2</sub> hybrid film and TiO<sub>2</sub> film. For TiO<sub>2</sub> film, only one F 1s peak at 684.3 eV was observed, which is attributed to F<sup>−</sup> adsorbed on the surface of TiO<sub>2</sub>. For MB/TiO<sub>2</sub> hybrid film, besides the peak at 684.2 eV, a new peak appeared at 685.7 eV. A positive shift of 1.5 eV in the F 1s binding energy of MB/TiO<sub>2</sub> hybrid film with respect to TiO<sub>2</sub> demonstrates that the chemical environment of fluorine is markedly different between the two systems. Therefore, in MB/TiO<sub>2</sub> hybrid film, MB should interact with the surface F site through the CN ring (as illustrated in Scheme 2).

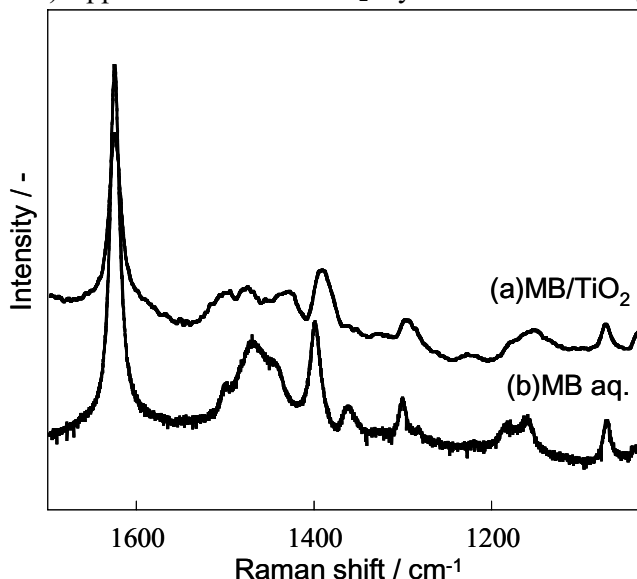


Figure 4. Raman spectra of: (a) MB/TiO<sub>2</sub> film and (b) MB in aqueous solution.

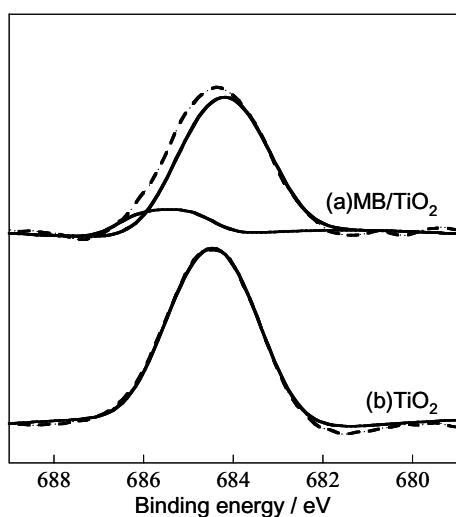
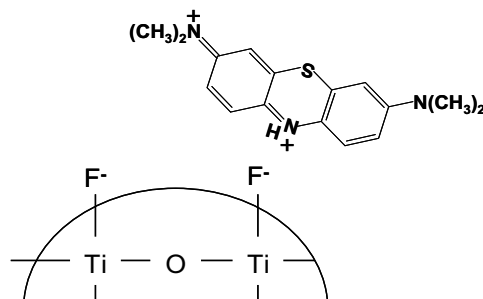


Figure 5. XPS spectra of F 1s: (a) MB/TiO<sub>2</sub> film and (b) TiO<sub>2</sub> film.



Scheme 2. Incorporation model of MB into TiO<sub>2</sub>.

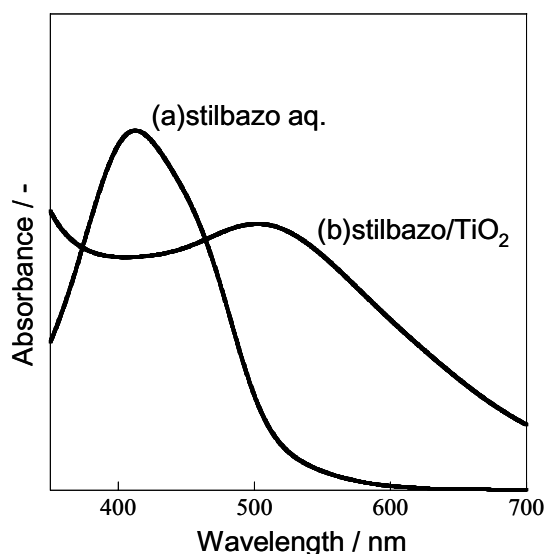
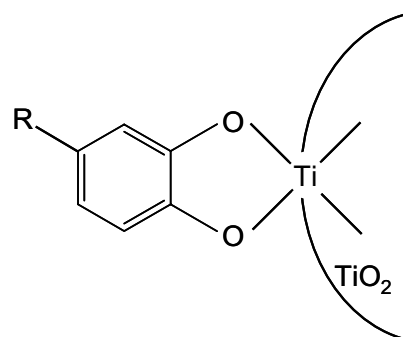


Figure 6. Absorption spectra of: (a) stilbazotriphenylamine in aqueous solution and (b) stilbazotriphenylamine/TiO<sub>2</sub> film.



Scheme 3. Incorporation model of stilbazotriphenylamine into TiO<sub>2</sub>.

Figure 6 shows the absorption spectra of free stilbazotriphenylamine in aqueous solution and stilbazotriphenylamine/TiO<sub>2</sub> hybrid film. For stilbazotriphenylamine/TiO<sub>2</sub> hybrid film, the absorption spectrum becomes broad and shifted to longer wavelength. This strong shift of the absorption maximum of about 90 nm from 410 to 500 nm upon incorporation indicates a very large electronic coupling between stilbazotriphenylamine and TiO<sub>2</sub>, and the appearance of a new ligand-to-metal charge transfer absorption band. We assume that there is a linkage of stilbazotriphenylamine through the two chelating -OH groups of its catechol moiety to one surface Ti<sup>4+</sup> ion with a bidentate linkage (as illustrated in Scheme 3). This is a favorable conformation of bond angles and distances for octahedrally coordinated surface of Ti atoms (16). It has been reported that TiO<sub>2</sub> can form a strong charge transfer complex with the catechol dye molecule such as alizarin (17,18).

Figure 7 shows IR spectra of free aluminon in aqueous solution and aluminon/TiO<sub>2</sub> hybrid film. C-O stretching vibration shifted toward lower wavenumber from 1257 cm<sup>-1</sup> to 1243 cm<sup>-1</sup> upon incorporation. This shift indicates an interaction of the -OH group with TiO<sub>2</sub>. The carboxylate bond could not be identified in aluminon/TiO<sub>2</sub> film. However, the degree of disassociation of the carboxyl group is higher than that of -OH group. Therefore, we assume that the carboxyl group bond to TiO<sub>2</sub>, and there is a linkage of aluminon through the chelating -OH group and carboxyl group to one surface Ti<sup>4+</sup> ion with a bidentate linkage as

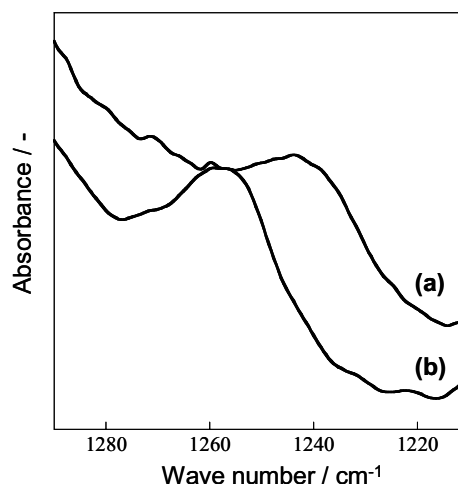


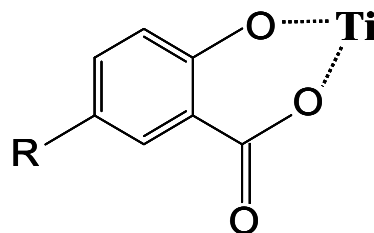
Figure 7. IR spectra of (a) aluminon/TiO<sub>2</sub> film and (b) aluminon in aqueous solution.



illustrated in Scheme 4 (19).

#### Characterization of organic dyes/metal oxide hybrid films

The density of dye molecules in the as-deposited stilbazo/TiO<sub>2</sub>, PCV/TiO<sub>2</sub> and MB/TiO<sub>2</sub> films were calculated according to Lambert-Beer's Law from the absorbance of the dye at the absorption maximum (Table 2). The density of stilbazo and PCV was higher than that of MB, although molecular size of stilbazo and PCV is larger than that of MB. This indicates that the affinity of catechol dye molecules (stilbazo and PCV) toward TiO<sub>2</sub> is better than that of MB.



Scheme 4. Incorporation model of aluminon into TiO<sub>2</sub>.

**Table 2.** The density of organic dyes in TiO<sub>2</sub> film.

| Sample                      | Density /<br>Molecules · nm <sup>-3</sup> |
|-----------------------------|---|
| MB / TiO <sub>2</sub>       | 0.074                                     |
| PCV / TiO <sub>2</sub>      | 0.973                                     |
| stilbazo / TiO <sub>2</sub> | 0.124                                     |

The relationship between concentration of dye solution and the amount of deposited Ti was studied by ICP (Figure 8). By incorporating MB, there is no change in the amount of deposited Ti in TiO<sub>2</sub> and MB/TiO<sub>2</sub> films. However, the amount of deposited Ti decreased in the case of catechol dye molecule as concentration of dye solution increases. Catechol dyes form chelate complex with TiF<sub>6</sub><sup>2-</sup> in treatment solution. Therefore, catechol dye molecules inhibit nuclear formation of TiO<sub>2</sub> and the amount of deposited Ti decreases.

Surface morphology of TiO<sub>2</sub> film and MB/TiO<sub>2</sub>, stilbazo/TiO<sub>2</sub> and PCV/TiO<sub>2</sub> hybrid films deposited on glass slides were studied by SEM (Figure 9). TiO<sub>2</sub> and MB/TiO<sub>2</sub> film show dense and compact substrate coverage. The particle size is ca. 40 nm in the TiO<sub>2</sub> and MB/TiO<sub>2</sub> samples. On the other hand, stilbazo/TiO<sub>2</sub> and PCV/TiO<sub>2</sub> films show markedly uneven. The particle size is ca. 160 nm in stilbazo/TiO<sub>2</sub> sample and ca. 90 nm in the PCV/TiO<sub>2</sub> sample. The particle size

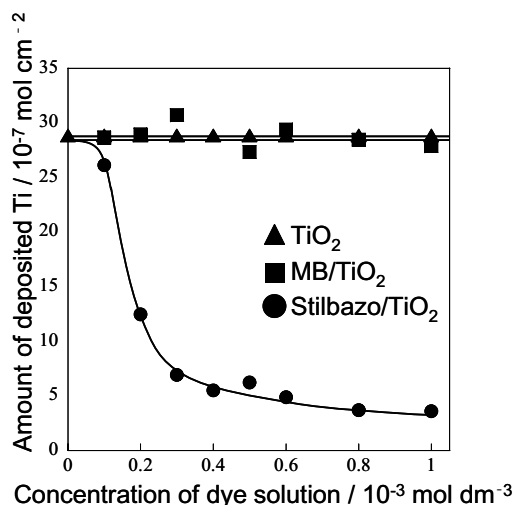


Figure 8. The relationship between concentration of dye solution and the amount of deposited Ti.

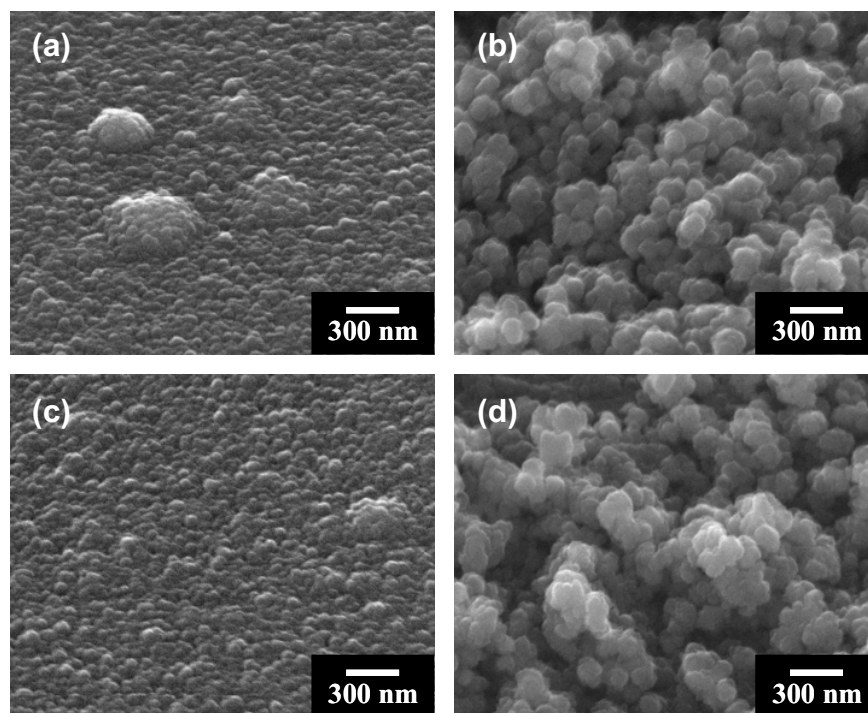


Figure 9. SEM images of: (a)  $\text{TiO}_2$ , (b) stilbazo/ $\text{TiO}_2$ , (c) MB/ $\text{TiO}_2$  and (d) PCV/ $\text{TiO}_2$  hybrid films.

in catechol dyes/ $\text{TiO}_2$  films is considerably larger than in  $\text{TiO}_2$  and MB/ $\text{TiO}_2$  films. In treatment solution, catecholate dye molecules form a charge transfer complex with  $\text{TiO}_2$  through the catechol moiety with a bidentate linkage. Therefore, catecholate dye molecules inhibit nuclear formation of  $\text{TiO}_2$ . As a result,  $\text{TiO}_2$  particles in catechol dyes/ $\text{TiO}_2$  samples grow larger. X-ray diffraction of  $\text{TiO}_2$ , MB/ $\text{TiO}_2$ , PCV/ $\text{TiO}_2$  and stilbazo/ $\text{TiO}_2$  films deposited on glass slides are shown in Figure 10. These diffraction peaks could be assigned to the anatase structure. It is well known that anatase is the  $\text{TiO}_2$  phase usually favored by LPD, due to the presence of titanium strongly complexing ions such as the fluoride (13,20). Preferential  $c$  axis orientation has been reported previously both for  $\text{TiO}_2$  films prepared by LPD, and has been attributed to the preferential adsorption of fluoride to planes parallel to this direction (21,22). On the other hand, the intensity of (1 0 1) reflection of the anatase increased in the case of catechol

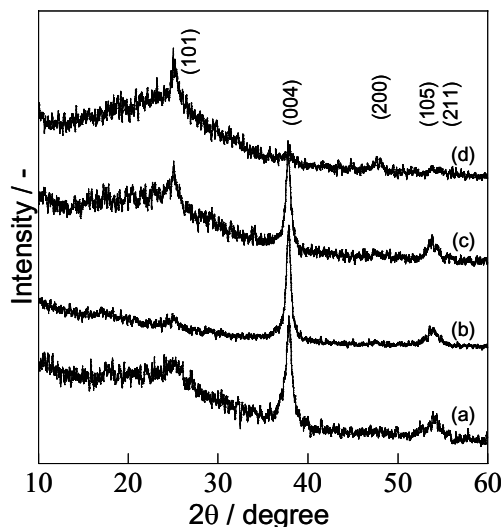


Figure 10. X-ray diffraction of: (a)  $\text{TiO}_2$ , (b) MB/ $\text{TiO}_2$ , (c) PCV/ $\text{TiO}_2$  and (d) stilbazo/ $\text{TiO}_2$  films.

dyes. It is caused by inhibition of nuclear formation of  $\text{TiO}_2$ .

### Conclusion

The organic compound/metal oxide composite thin films have been successfully prepared by the LPD method. The development of LPD technique has open a way for the preparation of organic/inorganic hybrid materials.

Due to the negative surface charge on  $\text{SiO}_2$ ,  $\text{TiO}_2$  and  $\text{ZrO}_2$ , hybrid films of cationic MB and NR were fabricated. However, hybrid films of cationic RB and MG did not show any color in  $\text{TiO}_2$  and  $\text{ZrO}_2$  systems due to its steric hindrance. Catecholate dye molecules such as stilbazo and PCV have a linkage through the two chelating -OH groups of its catechol moiety to one surface  $\text{Ti}^{4+}$  ion with a bidentate linkage. It is assumed that there is a linkage of aluminon through the chelating -OH group and carboxyl group to one surface  $\text{Ti}^{4+}$  ion with a bidentate linkage.

The density of stilbazo and PCV in metal oxide film was higher than that of MB. This is because the affinity of catecholate dye molecules (stilbazo and PCV) toward  $\text{TiO}_2$  is better than that of MB. There is no change in the amount of deposited Ti in  $\text{TiO}_2$  and MB/ $\text{TiO}_2$  films. However, the amount of deposited Ti decreased in the case of catecholate dye molecule. This implies that catechol dye forms chelate complex with  $\text{TiF}_6^{2-}$  in treatment solution and inhibits nuclear formation of  $\text{TiO}_2$ . The particle size of  $\text{TiO}_2$  in stilbazo/ $\text{TiO}_2$  and PCV/ $\text{TiO}_2$  films is considerably larger than in  $\text{TiO}_2$  and MB/ $\text{TiO}_2$  samples. In treatment solution, catecholate dye molecules inhibit nuclear formation of  $\text{TiO}_2$ .

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### References

1. D. A. Loy, *Mater. Res. Bull.*, **26**, 364 (2001).
2. D. O'Hare in *Inorganic materials*; D. W. Bruce, D. O'Hare, Eds; John Wiley & Sons: Chichester, Brisbane, Toronto, Singapore, p.166 (1992).
3. G. A. Ozin, *Adv. Mater.*, **4**, 612 (1992).
4. S. Y. Choi, M. Mamak, N. Coombs, N. Chopra and G. A. Ozin, *Nano Lett.*, **4**, 1231 (2004).
5. D. Gutiérrez-Tauste, I. Zumeta, E. Vigil, M. A. Hernández-Fenollosa, X. Domènech and J. A. Ayllón, *J. Photochem. Photobiol., A Chem.*, **175**, 165 (2005).
6. S. Deki, Y. Aoi, O. Hiroi and A. Kajinami, *Chem. Lett.*, **22**, 433 (1996).
7. S. Deki, Y. Aoi, Y. Miyake, A. Gotoh and A. Kajinami, *Mater. Res. Bull.*, **31**, 1399 (1996).
8. S. Deki, Y. Aoi and A. Kajinami, *J. Mater. Sci.*, **32**, 4269 (1997).
9. S. Deki, Y. Aoi, J. Okibe, H. Yanagimoto, M. Mizuhata and A. Kajinami, *J. Mater. Chem.*, **7**, 1769 (1997).
10. L. Li, M. Mizuhata and S. Deki, *Appl. Surf. Sci.*, **239**, 292 (2005).

11. M. Tatemichi, M. Sakamoto, M. Mizuhata, S. Deki and T. Takeuchi, *J. Am. Chem. Soc.*, **129**, 10906 (2007).
12. J. G. Yu, H. G. Yu, B. Cheng, X. J. Zhao, J. C. Yu and W. K. Ho, *J. Phys. Chem. B*, **107**, 13871 (2004).
13. D. Gutiérrez-Tauste, X. Domènech, M. A. Hernández-Fenollosa and J. A. Ayllón, *J. Mater. Chem.*, **23**, 2249 (2006).
14. H. Park and W. Choi, *J. Phys. Chem. B*, **108**, 4086 (2004).
15. D. Gutiérrez-Tauste, X. Domènech, N. Casan-Pastor and J. A. Ayllón, *J. Photochem. Photobiol., A Chem.*, **187**, 45 (2007).
16. T. Rajh, J. M. Nedeljkovic, L. X. Chen, O. Poluektov, M. C. Thurnauer, *J. Phys. Chem., B*, **103**, 3515 (1999).
17. G. Ramakrishna, Sandeep Verma, D. Amilan Jose, D. Krishna Kumar, Amitava Das, Dipak K. Palit and Hirendra N. Ghosh, *J. Phys. Chem. B*, **110**, 9012 (2006).
18. R. Huber, S. Sporlein, J. E. Moser, M. Gratzel and J. Wachtveitl, *J. Phys. Chem. B*, **104**, 8995 (2000).
19. K. D. Dobson and A. J. McQuillan, *Spectrochim. Acta A*, **56**, 557 (2000).
20. H. Pizem, C. N. Sukenik, U. Sampathkumaran, A. K. McIlwain and M. R. De Guire, *Chem. Mater.*, **14**, 2476 (2002).
21. H. Pizem, O. Gershevit, Y. Goffer, A. A. Frimer, C. N. Sukenik, U. Sampathkumaran, X. Milhet, A. McIlwain, M. R. De Guire, M. A. B. Meador and J. K. Sutter, *Chem. Mater.*, **17**, 3205 (2005).
22. A. M. Peiró, E. Vigil, J. Peral, C. Domingo, X. Domènech and J. A. Ayllón, *Thin Solid Films*, **411**, 185 (2002).